



## Supplementary Materials for

### **Dynamic interplay between metal nanoparticles and oxide support under redox conditions**

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#### **Other Supplementary Material for this manuscript includes the following:**

Movies S1 to S8

## **Materials and Methods**

### **Synthesis**

The catalyst was prepared by incipient wetness impregnation of a titanium (IV) oxide support (Aeroxide® P25, Acros Organics) with tetraammineplatinum (II) nitrate (99.995 %, Sigma Aldrich) dissolved in ultrapure MilliQ® water. After calcination at 200 °C in static air for 5 h (heating rate: 10 °C min<sup>-1</sup>), the catalyst with a platinum weight loading of 2% was transferred into a quartz tube and heated under He flow (50 mL min<sup>-1</sup>) in a tubular oven to 700 °C for 1 h (heating rate: 10 °C min<sup>-1</sup>). Subsequent XRD measurements showed no signs of anatase phase after calcination. A detailed synthesis protocol can be found elsewhere (5).

### **In situ TEM**

In situ TEM studies were performed using the aberration-corrected 300 kV JEOL Grand-ARM300F of the Scientific Center of Optical and Electron Microscopy (ScopeM) at ETH Zürich. A DENSsolutions Climate+ System was used to study samples under controlled gas-flow, pressure and temperature conditions. The setup was coupled to a quadrupole mass spectrometer (MS) (JEOL, JMS-TQ4000GC), for simultaneous detection of catalytic conversion. For additional experiments, the quadrupole mass spectrometer was exchanged with a (RC PIC Quadrupole Mass Spectrometer, Hiden Analytical). After dispersion in chloroform and ultrasonication for 30 minutes, sample material was drop-deposited onto the heating chip, and then mounted onto the DENSsolutions Climate holder. After leak-testing and insertion of the holder in the microscope, the samples were exposed to oxygen for 30 minutes at 300 °C in order to remove residual carbon contamination ( $p_{O_2} = 200$  mbar). This cleaning procedure prevents the building up of carbon deposits during subsequent electron beam imaging. In order to make sure that the observed changes were not caused by electron beam irradiation, we adjusted the illumination conditions throughout

the experiments such that the electron beam dose rate was below the threshold electron dose determined by Kuwauchi *et al.* (36) ( $\sim 1 \times 10^4 \text{ e}^- \text{ \AA}^{-2} \text{ s}^{-1}$ ). We furthermore compared regions that were exposed to the beam during gas-phase switching with other, non-exposed regions. Since all observed particles showed identical encapsulation (and reproducible behavior in multiple experiments), we can safely conclude that the state of the particles and overgrowth layer are not due to electron beam irradiation.

## **Protocol**

### Pretreatment

After the cleaning procedure, oxygen was removed from the gas mixture and the sample was exposed to a reductive treatment in order to induce a classical SMSI state (3,4,32). The pretreatment consisted of a heating step to 600 °C in helium followed by exposure to hydrogen ( $p_{\text{H}_2} = 700 \text{ mbar}$ ) for 60 minutes.

The final step of the pretreatment consisted of flushing out hydrogen with helium that was subsequently replaced by oxygen ( $p_{\text{O}_2} = 700 \text{ mbar}$ ) in order to prepare the system for the study of redox catalysis under oxygen rich conditions.

### Initiation of the classical SMSI state

The initial Pt/TiO<sub>2</sub> (2 weight % of Pt) contained Pt particles of various shapes in the size range between 2 and 20 nm. Amongst them, the most representative particles were about 10-15 nm in size and shaped in the form of a truncated cuboctahedron, exposing mostly {111} and {100} facets (see Fig. **S1 A-C**). The starting point of our investigation was the classical SMSI state, which was induced by heating the catalyst to 600 °C in a flow of hydrogen at 700 mbar inside the TEM using a DENSolutions Climate holder. This in situ reductive activation results in the typical

encapsulation of Pt particles by a thin, partially reduced layer of TiO<sub>2</sub> (5–9). Once this classical SMSI state was established (**Fig. S1 D**), the gas phase was switched in steps, first from reducing conditions to inert He, then to pure O<sub>2</sub> and finally, to an oxygen-rich, reactive H<sub>2</sub>+O<sub>2</sub> mixture. During the gas-switching, the sample temperature and total pressure were kept constant at 600 °C and 760 mbar, respectively.

#### Switching to oxygen atmosphere

No structural changes were observed during the gradual replacement of hydrogen by helium (**Fig. S2**). After hydrogen was fully flushed out of the reactor, O<sub>2</sub> was introduced. The switching from inert to oxidizing conditions did not show any obvious effect on the particle shape but resulted in a change of the overlayer structure (panels **D** and **E** in **Fig. S1**). It gained in thickness and developed a particular double-layered arrangement. The swelling of the overlayer and the frequently observed formation of patches of excess material (see arrows in **Fig. S1 F** & **Fig. S2 C**) are indicative of a re-oxidation of the initial SMSI overlayer, possibly with contributions from dealloying of a platinum-titanium alloy that formed at the particle surface during high-temperature reduction (5). **Fig. S1 E** shows how the configuration of the overlayer under oxygen depends on the surface structure of the Pt NPs: ordered structures form on the {111} facets, and ill-structured, amorphous titania exist on the {100} planes and {110} microfacets as well as kinks and edges between these facets. Ordered low-energy configurations can only form on facets where the structural incoherence between possible crystalline configurations of the overlayer and the platinum surface and thus, the interfacial energy, is low (75). Since the structures of the overlayer and the composition of the interface depend on the chemical environment, an integral description of SMSI layers, especially on the basis of ex situ observations, is insufficient. This becomes even

more evident in the next step, when we describe the observations made upon switching from oxidizing conditions to the reactive redox regime.

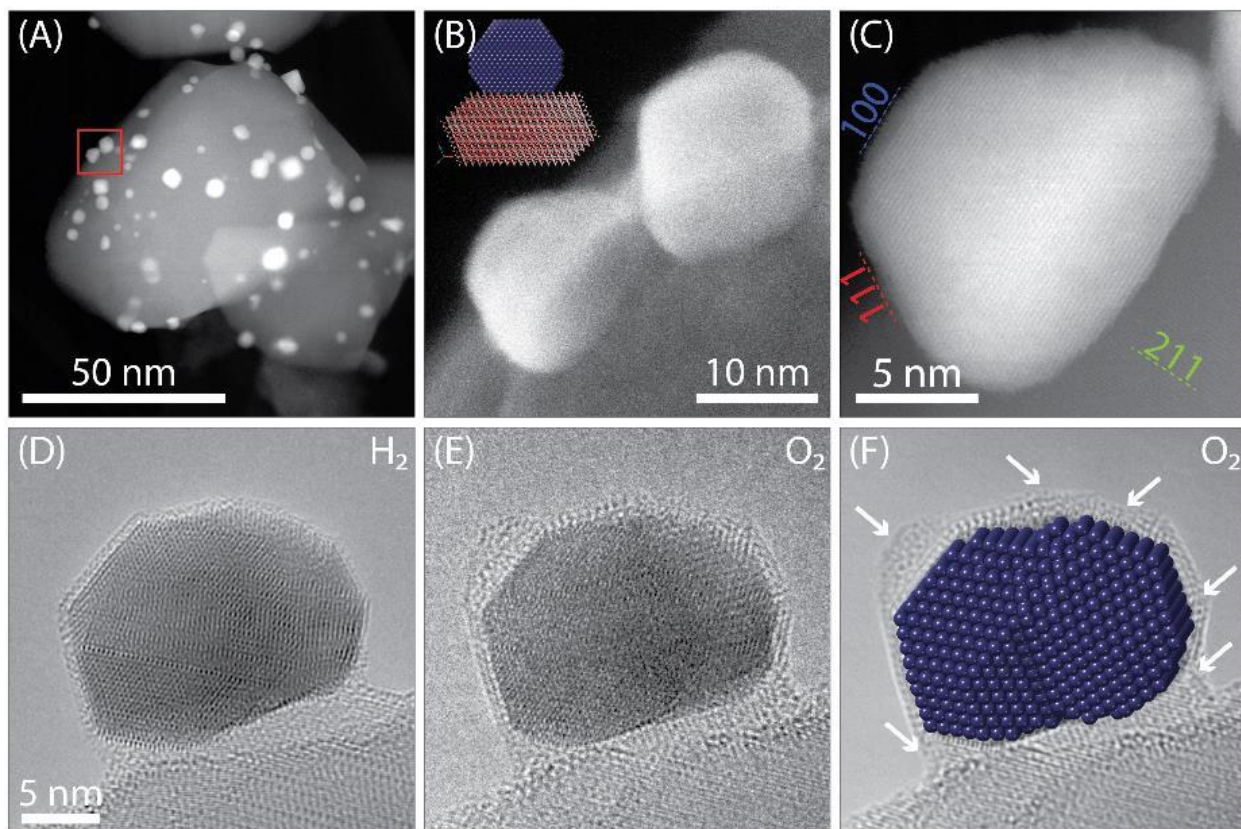
#### Reaction conditions

After 30 minutes, a small amount of H<sub>2</sub> ( $p_{\text{H}_2} = 60$  mbar) was added to the oxygen flow. The O<sub>2</sub> partial pressure was held constant over the course of the experiment. When H<sub>2</sub> was retracted from the gas mixture, the resulting partial pressure loss was compensated by adding the same amount of helium to the system.

#### Water addition

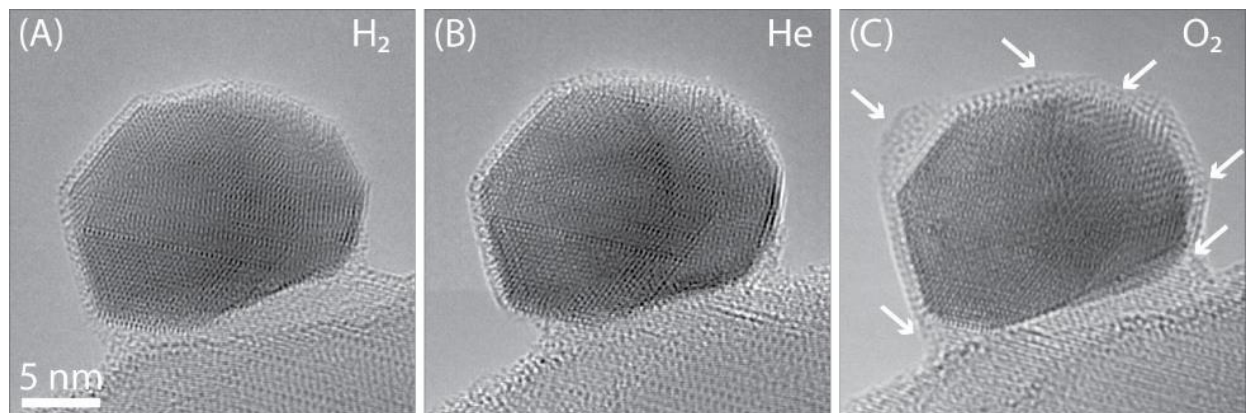
Additional experiments were performed in order to elucidate the influence of water on the observed dynamics. To saturate the feed gas flow with water vapor, a capillary humidifier was designed and integrated into the gas-feeding system of the *in situ* TEM set-up. The saturator consists of a P80 polypropylene tubular membrane (Zena-Membranes) that is submerged in degassed ultrapure water and sealed against surrounding atmosphere. The pore size and the hydrophobic character of the membrane prevent water diffusion into the inside lumen of the membrane but allow for water vapor transport. The theoretical upper limit for the water vapor partial pressure in the system was calculated based on the saturation vapor pressure of water at 23°C and the relatively lower pressure in the nanoreactor (1000 mbar) with respect to that of the humidifier (1414 mbar) that is situated upstream the nanoreactor. For our experiments, the maximal excess water vapor that could be added was 19.8 mbar. Note that due to the replacement of a defective mass-flow-controller, the hydrogen flow required to enter the redox regime differed slightly from previous experiments (set value of  $x_{\text{H}_2} = 0.14$  instead of 0.08).

**Fig. S1.**



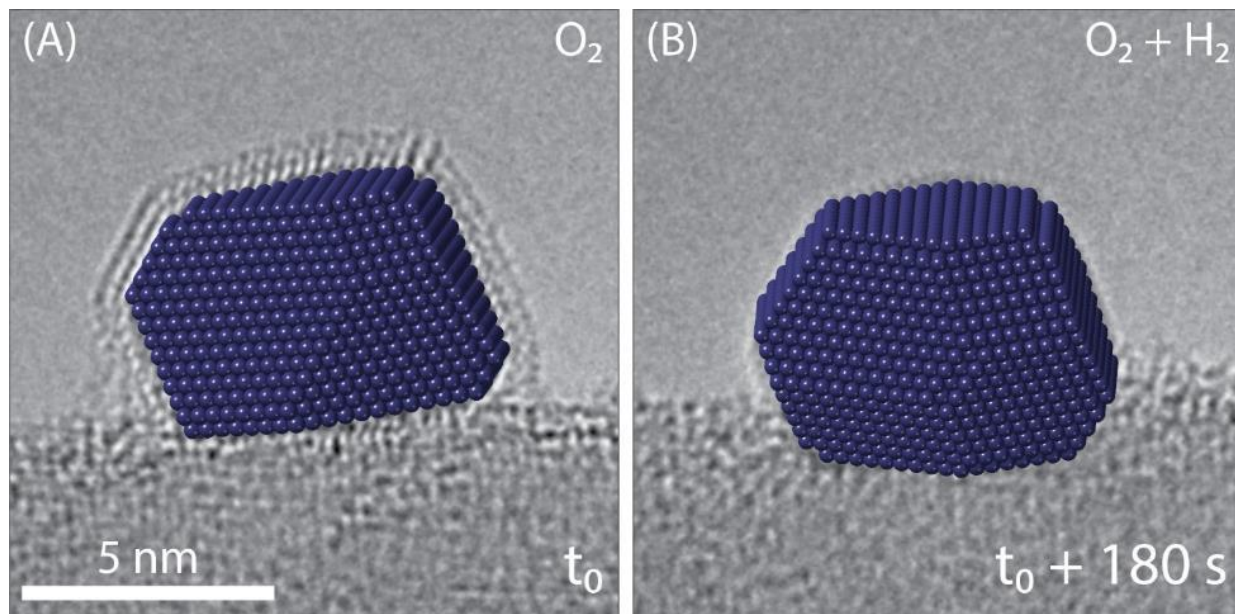
**Fig. S1. Starting material and encapsulated Pt NP under H<sub>2</sub> and O<sub>2</sub>.** **A:** Annular dark-field (ADF) STEM image of the Pt/TiO<sub>2</sub> catalyst, showing the distribution of small Pt particles on larger TiO<sub>2</sub> support particles. **B:** Secondary electron image of two representative Pt particles that show a low surface wetting and are shaped in the form of truncated cuboctahedra. **C:** ADF STEM image recorded at higher magnification revealing lattice fringes and the orientation of exposed platinum surfaces. **D:** A platinum particle with the classical SMSI overlayer imaged at 600 °C in 700 mbar hydrogen. **E:** The same particle after switching from hydrogen, via He to an O<sub>2</sub> atmosphere (600 °C, p<sub>O<sub>2</sub></sub> = 700 mbar). **F:** Image generated by summing over 30 subsequent frames (starting with **E**), corresponding to an integral recording time of 5s and overlaid with a structural model of platinum (not to scale).

**Fig. S2.**



**Fig. S2.** Platinum-titania at 600 °C under **A:** 1 bar of H<sub>2</sub>, **B:** 1 bar of He and **C:** 700 mbar of O<sub>2</sub>. Images integrated over 5 seconds exposure time respectively. Arrows in **C** indicate amorphous patches of the overlayer.

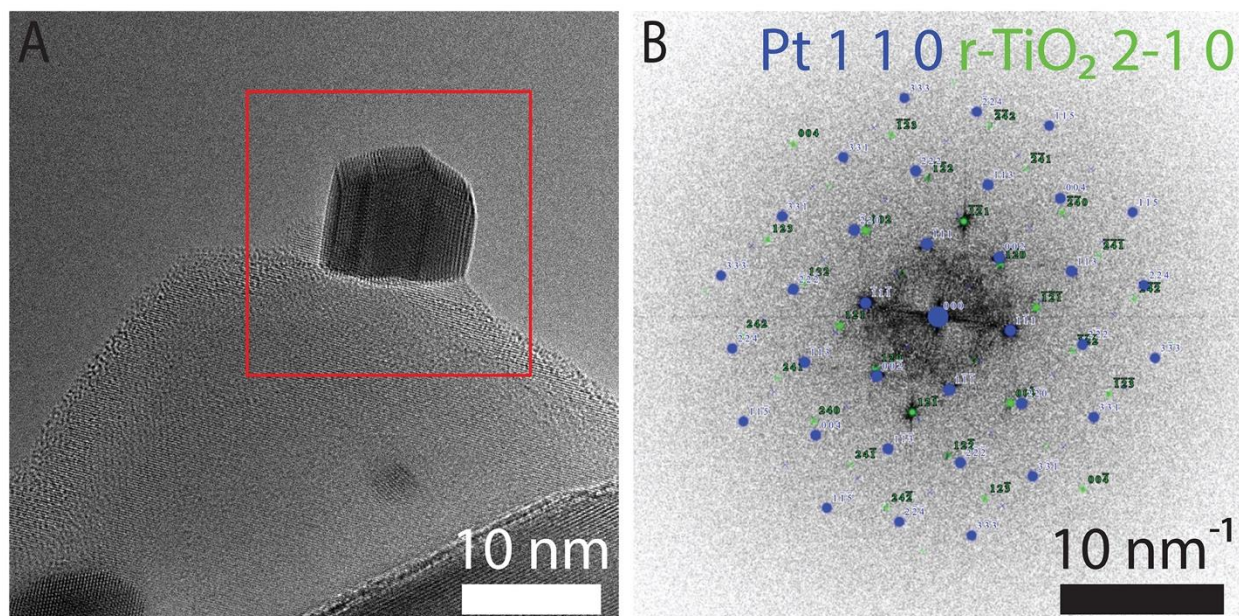
**Fig. S3.**



**Fig. S3. Pt particle on TiO<sub>2</sub> at 600 °C.** (A) under 700 mbar of O<sub>2</sub> and (B) under 700 mbar of O<sub>2</sub> and 60 mbar of H<sub>2</sub>. The frames shown in (A) and (B) correspond to **Fig 1 A** and **F** respectively. Superimposed particle models show a change in surface wetting and slight expansion of {100} facets.

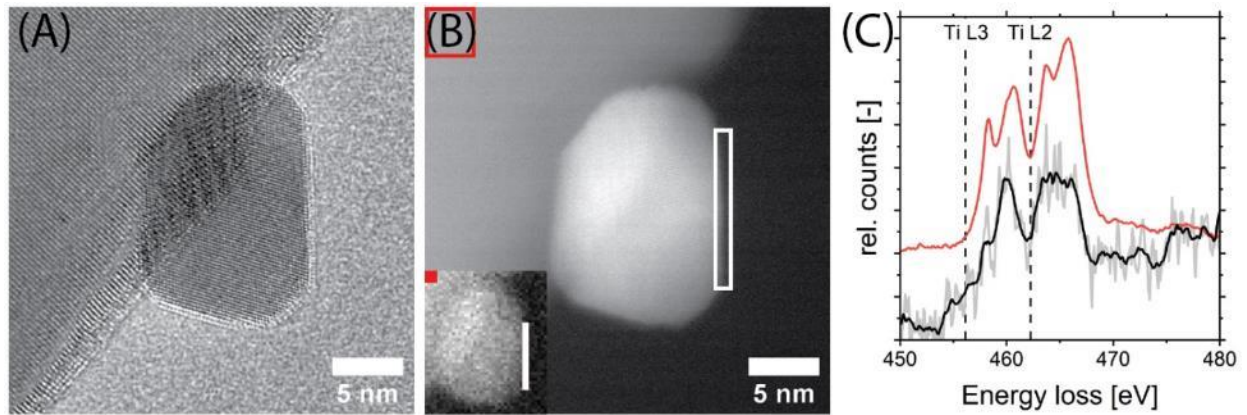


**Fig. S4.**



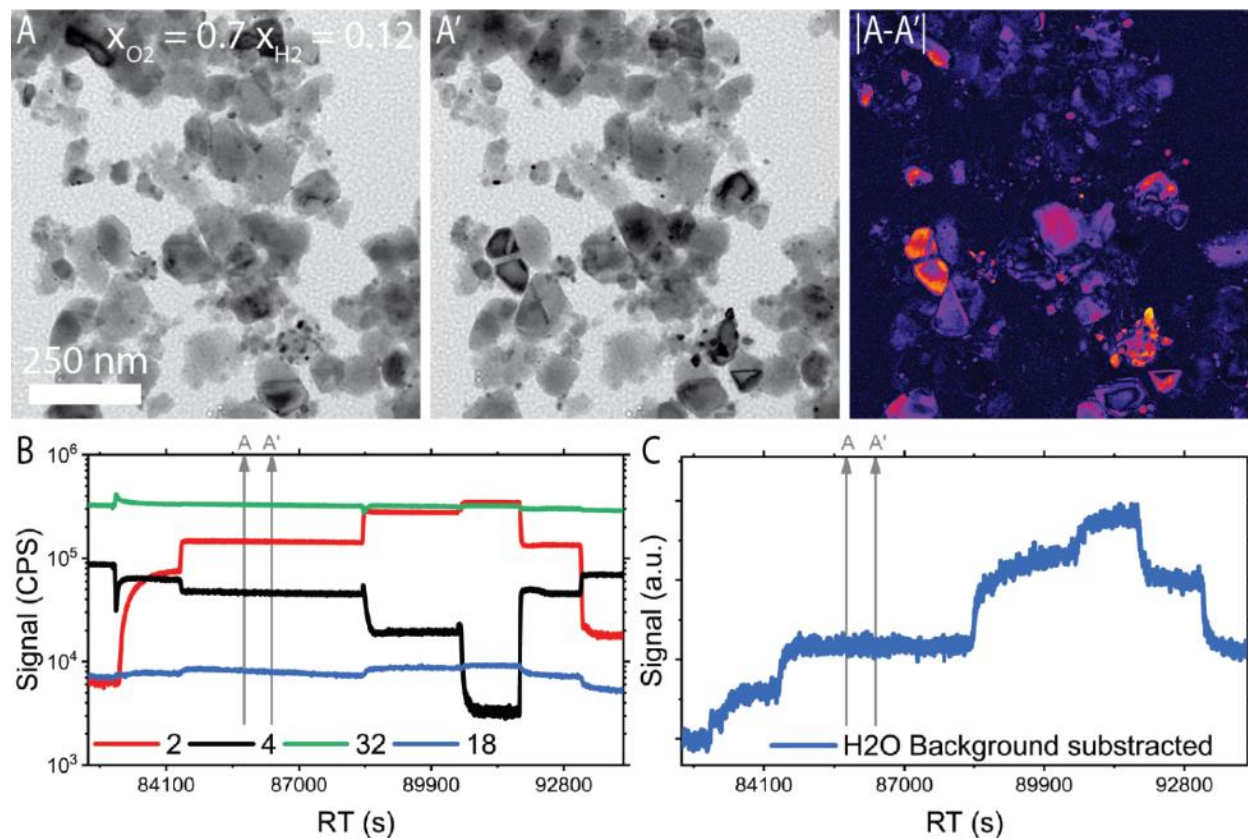
**Fig. S4. Fast Fourier Transform (FFT) analysis of ROI in Fig. 3 E. (A)** Micrograph shown in **Fig. 3 E**. Red square marks area from where FFT shown in **(B)** was extracted. **(B)** Indexed FFT. Viewing direction for Pt and TiO<sub>2</sub> is shown in legend. Fourier transform in **(B)** shows a proximity in spacing and orientation of Pt {220} and TiO<sub>2</sub> {002} planes and similarly, of Pt {002} and TiO<sub>2</sub> {120} planes, indicating a preferential orientation due to minimization of the lattice misfit induced interfacial strain (76).

**Fig. S5.**



**Fig. S5. Pt/TiO<sub>2</sub> at 600 °C and 700 mbar of O<sub>2</sub>.** (A) HRTEM image of encapsulated platinum particle. (B) The same particle imaged with HAADF-STEM. Inset shows the recorded spectral image of the same region. (C) Extracted electron energy loss spectra (EELS) from the regions indicated in (B) by a red square and a white bar. Ti L<sub>3,2</sub> edges are clearly visible for both, the support and the particle edge.

**Fig. S6.**



**Fig. S6. Mass conversion data.** Mass conversion data of the platinum titania catalyst under a constant 700 mbar O<sub>2</sub> pressure and 0, 60, 120, 300, 0 mbar of H<sub>2</sub> at 600°C. (A) in situ TEM micrograph of the Pt/TiO<sub>2</sub> sample under 700 mbar of O<sub>2</sub> and 120 mbar of H<sub>2</sub> at 600°C. (A'): the same ROI 10 minutes later. |A-A'|: Color-coded difference image of (A) and (A'). Bright areas indicate particle migration. (B) Raw MS response, (C) Magnified water signal, treated additionally by subtracting a linear background. In (B) and (C), the timestamp of (A) and (A') is indicated by grey arrows.

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